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New compounds $\text{Li}_3\text{Ba}_2\text{Bi}_3(\text{XO}_4)_8$ ($X = \text{Mo}, \text{W}$): synthesis and properties

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Abstract

New compounds $\text{Li}_3\text{Ba}_2\text{Bi}_3(\text{XO}_4)_8$ ($X = \text{Mo}, \text{W}$) were obtained by the ceramic technology. Those are the first representatives of the ternary molybdates and tungstates $\text{Li}_3\text{Ba}_2\text{R}_3(\text{XO}_4)_8$ family, which contain different from the rare earth elements trivalent metal. The sequence of chemical transformations occurring during the $\text{Li}_3\text{Ba}_2\text{Bi}_3(\text{WO}_4)_8$ formation has been established. The primary characterization of the obtained phases was carried out and their ion-conducting properties were studied. The synthesized compounds are shown to melt incongruently, isostructural to the lanthanide-containing analogues (structural type of $\text{BaNd}_2(\text{MoO}_4)_4$, sp. gr. $C2/c$) and crystallize in the monoclinic crystal system with unit cell parameters $a = 5.2798(1)$, $b = 12.8976(4)$, $c = 19.2272(5)$ Å, $\beta = 90.978(2)^\circ$ ($X = \text{Mo}$), $a = 5.2733(2)$, $b = 12.9032(4)$, $c = 19.2650(6)$ Å, $\beta = 91.512(3)^\circ$ ($X = \text{W}$). $\text{Li}_3\text{Ba}_2\text{Bi}_3(\text{XO}_4)_8$ are found to undergo the diffuse first-order phase transitions at 441°C (molybdate) and 527°C (tungstate), after that their conductivity reaches values of 10^{-3} – 10^{-4} S/cm.

Keywords: triple molybdates and tungstates, solid-state synthesis, powder X-ray diffraction study, thermal properties, ionic conductivity

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1. Introduction

Currently, the class of ternary molybdates includes more than 700 representatives, characterized by a large stoichiometric and structural diversity, and belongs to the most dynamically replenished groups of complex oxide compounds containing a tetrahedral anion and three different cations. Triple molybdates of different valence metals not only have a high material science potential, but also due to the wide possibilities of varying the elemental and quantitative compositions are convenient model objects for establishing genetic relationships in the series of composition – structure – properties of compound – properties of materials. A significant place among them is occupied by the family of triple molybdates obtained for all lanthanides and yttrium with the composition $\text{Li}_3\text{Ba}_2\text{R}_3(\text{MoO}_4)_8$, belonging to the structure type $\text{BaNd}_2(\text{MoO}_4)_4$ (sp. gr. $C2/c$, $Z = 2$) and related to scheelite. These compounds are shown to possess promising luminescent and generation properties [1–3], as well as the properties of solid electrolytes [1, 4]. In particular, the obtained results aimed at study of $\text{Li}_3\text{Ba}_2\text{R}_3(\text{MoO}_4)_8$ ($R = \text{La, Gd, Y}$) doped with Eu^{3+} , Tb^{3+} , Er^{3+} , Nd^{3+} , indicate the prospects for their use as new photo- and IR-luminophores and laser materials [2, 5–7]. Since 2009, studies on the preparation of ternary tungstates of lithium-barium-lanthanides, isoformular and isostructural to the $\text{Li}_3\text{Ba}_2\text{Ln}_3(\text{WO}_4)_8$ which are, like them, are of great not only scientific but also practical interest started to appear [8–17]. Sizes and quality of grown crystals $\text{Li}_3\text{Ba}_2\text{Ln}_3(\text{WO}_4)_8$ ($\text{Ln} = \text{La, Gd, Y}$) [9–12, 18] doped with Nd^{3+} , Eu^{3+} , Tm^{3+} and other ions, allowed proceeding to a detailed study of the optical-generation characteristics of these new highly efficient laser media. Ceramics $\text{Li}_3\text{Ba}_2\text{La}_3(\text{WO}_4)_8$: Eu^{3+} [8] and $\text{Li}_3\text{Ba}_2\text{Gd}_3(\text{WO}_4)_8$: Tb^{3+} [15] can be used as red and green luminophores, respectively.

In this study, the first representatives of the considered family of phases containing in their composition a different from the rare earth trivalent element – ternary bismuth – containing molybdate and tungstate of composition $\text{Li}_3\text{Ba}_2\text{Bi}_3(\text{XO}_4)_8$ ($X = \text{Mo, W}$) were obtained by directed solid state synthesis. The primary characterization of the obtained compounds was carried out and their electrophysical properties were studied.

2. Experimental

Industrial reagents Li_2MoO_4 , Li_2WO_4 , XO_3 ($X = \text{Mo, W}$), Bi_2O_3 , BaMoO_4 , BaCO_3 (chemically pure) were used as source components for the synthesis of $\text{Li}_3\text{Ba}_2\text{Bi}_3(\text{XO}_4)_8$ ($X = \text{Mo, W}$). BaWO_4 was obtained by annealing of the stoichiometric mixture of BaCO_3 and WO_3 (600–850 °C, 70 h), $\text{Bi}_2(\text{MoO}_4)_3$ – by the reaction: $\text{Bi}_2\text{O}_3 + 3\text{MoO}_3 = \text{Bi}_2(\text{MoO}_4)_3$ (450–500 °C, 50 h). Tertiary bismuth tungstate does not exist; it could not be obtained by the solid-state method, as it was shown in the literature data [19] and proved by our unsuccessful attempts to synthesize those. Therefore, in this case the source component was an oxide mixture of Bi_2O_3 and WO_3 . AXO_4 ($A = \text{Ca, Sr, Cd, Pb}$; $X = \text{Mo, W}$), required to study of the possibility of realizing the considered structure in ternary molybdates and bismuth tungstates with complete or partial substitution of barium by another doubly charged cation, were obtained by the interaction of ACO_3 (chemically pure and analytical grade) and XO_3 by the reaction $\text{ACO}_3 + \text{XO}_3 = \text{AXO}_4 + \text{CO}_2$. The synthesis conditions were as follows: in case of Ca, Pb – 500–650 °C, Sr – 500–750 °C, Cd – 450–500 °C for 50–60 h; tungstates: Ca, Sr – 600–900 °C, Cd – 500–650 °C, Pb – 500–750 °C for 70–80 h. The single-phase of the synthesized materials was monitored by powder X-ray diffraction analysis. The obtained compounds were identified by comparison with the ICDD PDF-2 database [20].

Powder X-ray diffraction analysis (XRD) was performed using a Bruker D8 ADVANCE diffractometer ($\lambda\text{CuK}\alpha$, secondary monochromator, scanning step 0.02076°). Unit cell parameters of polycrystalline samples $\text{Li}_3\text{Ba}_2\text{R}_3(\text{XO}_4)_8$ ($X = \text{Mo, W}$) was calculated by the selection of an isostructural compound. Unit cell parameters were refined by the least squares method using the ICDD software package for preparation of the experimental standards. The Smith–Snyder F_{30} criterion was used as a validation criterion for X-ray patterns indexing [21].

Differential scanning calorimetry studies were carried out using NETZSCH STA 449C synchronous thermal analyser, $V_{\text{heat(cool)}} = 10^\circ/\text{min}$.

For the ion-conducting properties investigation the ceramic discs $\text{Li}_3\text{Ba}_2\text{Bi}_3(\text{XO}_4)_8$ ($X = \text{Mo, W}$) were prepared by pressing the powder at 1 kbar and annealing at 680 ($X = \text{Mo}$) or 730 °C ($X = \text{W}$) for 4 hours. The density of the obtained tablets was 90–95 % of the theoretical values. The

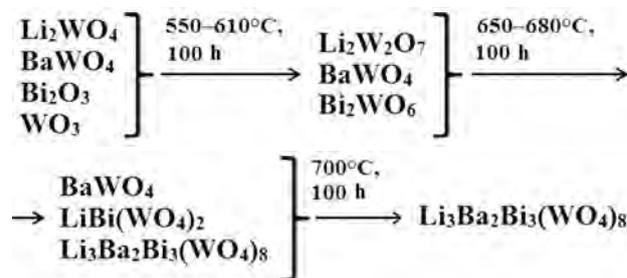
disks were in diameter of 10 mm and thickness of 1.8 mm. In order to prepare electrodes, the surfaces of the disks were coated with colloidal platinum, followed by annealing at 660 ($X = \text{Mo}$) or 710 °C ($X = \text{W}$) for 1 hour. The electrical conductivity measurements of the samples were tested using an impedance meter “Z-1500J” at selected frequencies from 1 Hz to 1 MHz in the temperature range of 200–650 °C ($X = \text{Mo}$) and 300–700 °C ($X = \text{W}$) with heating and cooling rates of 2 deg./min.

3. Results and discussion

In the single-phase polycrystalline state, the triple molybdate $\text{Li}_3\text{Ba}_2\text{Bi}_3(\text{MoO}_4)_8$ synthesized by annealing of stoichiometric mixtures of Li_2MoO_4 , BaMoO_4 , and $\text{Bi}_2(\text{MoO}_4)_3$ at 450–550 °C for 150 h, analogous triple tungstate $\text{Li}_3\text{Ba}_2\text{Bi}_3(\text{WO}_4)_8$ – 300 hour annealing of Li_2WO_4 , BaWO_4 , Bi_2O_3 and WO_3 , taken in a molar ratio of 3: 4: 3: 9, at 550–700 °C (intermediate homogenization was carried out every 15–20 h).

According to powder XRD data (Fig. 1), the sequence of chemical transformations occurring during the formation of $\text{Li}_3\text{Ba}_2\text{Bi}_3(\text{WO}_4)_8$ from a

stoichiometric mixture of oxides and tertiary tungstates, can be illustrated by the scheme [22]:



The formation of $\text{Li}_3\text{Ba}_2\text{Bi}_3(\text{MoO}_4)_8$ most likely also proceeds through the stage of formation of a double lithium-bismuth compound, but due to the close temperature ranges of the formation of intermediate and final products, the appearance of $\text{LiBi}(\text{MoO}_4)_2$ in the reaction mixture was not recorded.

According to the differential scanning calorimetry (DSC) data, the obtained compounds melt incongruently at 756 ($X = \text{Mo}$) and 786 °C ($X = \text{W}$). In addition to BaWO_4 and $\text{LiBi}(\text{WO}_4)_2$ and Bi_2WO_6 the presence of BaMoO_4 and $\text{LiBi}(\text{MoO}_4)_2$, tungstate was revealed in the cooled melt of molybdate by XRD analysis.

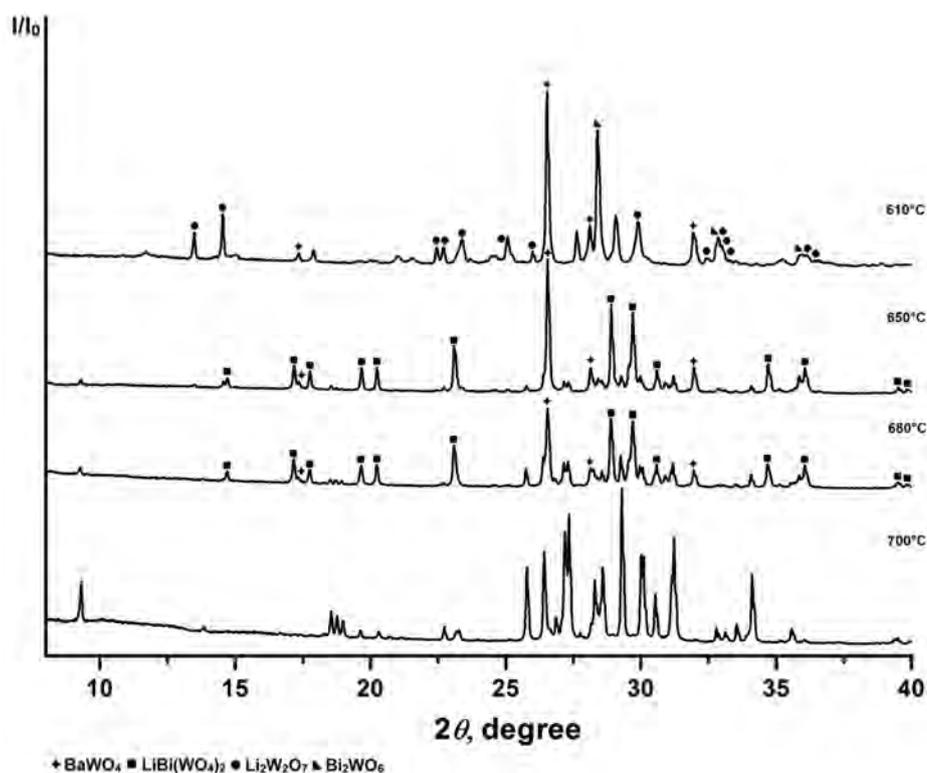


Fig 1. Powder X-ray diffraction patterns of the reaction mixture $\text{Li}_2\text{WO}_4 + 4\text{BaWO}_4 + 3\text{Bi}_2\text{O}_3 + 9\text{WO}_3$, sequentially annealed at different temperatures

Powder XRD patterns of $\text{Li}_3\text{Ba}_2\text{Bi}_3(\text{XO}_4)_8$ ($X = \text{Mo}, \text{W}$) were indexed satisfactory under the assumption of isostructurality to lanthanide-containing analogues (in the case of molybdate $F(30) = 217.1$ (0.0035; 39), tungstate $-F(30) = 162.3$ (0.0047; 39)). The obtained crystallographic characteristics are shown in Table 1, the results of indexing of $\text{Li}_3\text{Ba}_2\text{Bi}_3(\text{WO}_4)_8$ are shown in Table 2.

The possibility of realizing a similar structure in ternary bismuth molybdates and tungstates by replacing barium with another double-charged cation A^{2+} was investigated. However, attempts to synthesize $\text{Li}_3\text{A}_2\text{Bi}_3(\text{XO}_4)_8$ ($A = \text{Ca}, \text{Sr}, \text{Cd}, \text{Pb}$) were unsuccessful. The compositions $\text{Li}_3\text{Ba}_{1.9}\text{A}_{0.1}\text{Bi}_3(\text{MoO}_4)_8$ were obtained by the partial substitution of barium with strontium, cadmium,

Table 1. Crystallographic characteristics of $\text{Li}_3\text{Ba}_2\text{Bi}_3(\text{XO}_4)_8$ ($X = \text{Mo}, \text{W}$), sp. gr. $C2/c$, $Z = 2$

Compound	Unit cell parameters				$V, \text{\AA}^3$
	$a, \text{\AA}$	$b, \text{\AA}$	$c, \text{\AA}$	$\beta, ^\circ$	
$\text{Li}_3\text{Ba}_2\text{Bi}_3(\text{MoO}_4)_8$	5.2798(1)	12.8976(4)	19.2272(5)	90.978(2)	1309.12
$\text{Li}_3\text{Ba}_2\text{Bi}_3(\text{WO}_4)_8$	5.2733(2)	12.9032(4)	19.2650(6)	91.512(3)	1310.38

Table 2. Indexing results of powder X-ray diffraction pattern for $\text{Li}_3\text{Ba}_2\text{Bi}_3(\text{WO}_4)_8$

$2\theta_{\text{exp}}, ^\circ$	I/I_0	$d_{\text{exp}}, \text{\AA}$	h	k	l	$\Delta = 2\theta_{\text{exp}} - 2\theta_{\text{calc}}, ^\circ$	$2\theta_{\text{exp}}, ^\circ$	I/I_0	$d_{\text{exp}}, \text{\AA}$	h	k	l	$\Delta = 2\theta_{\text{exp}} - 2\theta_{\text{calc}}, ^\circ$
9.191	31	9.6140	0	0	2	-0.014	35.032	1	2.5593	-2	0	2	-0.002
13.723	3	6.4475	0	2	0	-0.009	35.293	1	2.5410	-1	3	5	-0.004
16.539	2	5.3555	0	2	2	-0.013	35.447	16	2.5303	0	2	7	-0.006
18.163	2	4.8802	1	1	0	+0.001	35.527	1	2.5248	2	0	2	-0.008
18.417	18	4.8134	0	0	4	-0.004	35.900	2	2.4994	1	3	5	-0.003
18.637	14	4.7571	-1	1	1	-0.004	36.810	1	2.4397	2	2	0	-0.005
18.857	11	4.7021	1	1	1	-0.004	37.084	1	2.4223	-1	1	7	+0.004
19.501	5	4.5482	0	2	3	-0.010	37.322	1	2.4074	0	0	8	+0.001
20.184	5	4.3958	-1	1	2	-0.002	38.601	1	2.3305	-1	3	6	-0.014
20.592	1	4.3097	1	1	2	-0.005	38.814	1	2.3182	-1	5	0	+0.007
22.601	10	3.9309	-1	1	3	-0.003	39.056	1L	2.3044	-1	5	1	+0.000
23.032	6	3.8583	0	2	4	-0.002	39.168	2	2.2981	1	5	1	+0.000
23.148	8	3.8392	1	1	3	-0.004	39.268	3	2.2924	1	3	6	-0.006
25.650	66	3.4701	-1	1	4	+0.002	39.382	5	2.2861	2	0	4	-0.012
26.298	57	3.3861	1	1	4	-0.001	39.819	2	2.2620	2	2	3	-0.010
26.723	16	3.3332	-1	3	0	0.005	39.945	2	2.2551	0	2	8	-0.005
26.943	9	3.3065	0	2	5	-0.005	41.030	1	2.1980	-2	2	4	-0.004
27.059	86	3.2926	-1	3	1	-0.004	41.228	1	2.1879	-1	5	3	-0.006
27.219	100	3.2736	1	3	1	-0.009	41.384	2	2.1800	-1	1	8	-0.003
27.639	8	3.2248	0	4	0	-0.009	41.554	1L	2.1714	1	5	3	-0.011
28.032	11	3.1804	0	4	1	-0.009	41.879	2	2.1553	2	2	4	+0.000
28.172	51	3.1650	-1	3	2	-0.006	41.975	2	2.1506	0	6	0	+0.002
28.464	45	3.1331	1	3	2	+0.000	42.245	1L	2.1375	0	6	1	+0.006
29.167	97	3.0592	-1	1	5	-0.007	42.946	2	2.1042	1	3	7	-0.005
29.880	63	2.9878	1	1	5	-0.003	43.041	1	2.0998	0	6	2	+0.021
29.982	28	2.9779	-1	3	3	-0.002	43.083	3	2.0979	-1	5	4	-0.002
30.402	26	2.9377	1	3	3	-0.001	43.158	1	2.0944	0	4	7	+0.024
31.001	47	2.8823	0	4	3	-0.001	43.378	2	2.0843	-2	2	5	-0.001
31.097	61	2.8736	0	2	6	-0.001	44.391	25	2.0390	0	6	3	-0.003
32.998	8	2.7123	-1	1	6	+0.000	44.584	23	2.0306	0	2	9	-0.008
33.411	14	2.6797	0	4	4	-0.003	44.740	2	2.0239	2	4	1	-0.033
33.781	5	2.6512	1	1	6	-0.012	45.180	12	2.0052	-2	4	2	+0.005
33.985	61	2.6357	2	0	0	+0.000	45.395	2	1.9962	-1	5	5	+0.001

End of Table 2

$2\theta_{\text{exp}}, ^\circ$	I/I_0	$d_{\text{exp}}, \text{\AA}$	h	k	l	$\Delta =$ $=2\theta_{\text{exp}} - 2\theta_{\text{calc}}, ^\circ$	$2\theta_{\text{exp}}, ^\circ$	I/I_0	$d_{\text{exp}}, \text{\AA}$	h	k	l	$\Delta =$ $=2\theta_{\text{exp}} - 2\theta_{\text{calc}}, ^\circ$
45.581	15	1.9885	2	4	2	+0.001	56.460	8	1.6285	-2	2	9	-0.006
45.881	3	1.9762	1	5	5	+0.009	56.557	2	1.6259	-3	3	0	-0.029
46.136	21	1.9659	-2	2	6	+0.004	56.627	11	1.6241	3	1	4	+0.009
46.190	26	1.9637	0	6	4	+0.004	56.810	4	1.6193	-2	6	3	-0.006
46.363	25	1.9568	-2	4	3	+0.003	56.869	7	1.6177	3	3	1	+0.006
46.948	22	1.9338	2	4	3	+0.001	57.142	6	1.6106	-3	3	2	+0.009
47.077	2	1.9288	0	4	8	-0.014	57.309	4	1.6063	2	6	3	+0.000
47.147	9	1.9261	0	0	10	+0.006	57.464	8	1.6024	-3	1	5	+0.002
47.311	17	1.9198	2	2	6	-0.006	57.651	4	1.5976	3	3	2	+0.003
48.018	6	1.8931	-2	4	4	+0.001	57.965	5	1.5897	2	2	9	+0.002
48.116	4	1.8895	-1	5	6	+0.000	58.112	2	1.5860	-3	3	3	-0.001
48.447	3	1.8774	0	6	5	-0.008	58.229	7	1.5831	-2	6	4	+0.013
48.680	5	1.8689	1	5	6	+0.003	58.287	14	1.5817	-1	5	9	+0.011
48.771	7	1.8657	2	4	4	+0.004	58.557	2	1.5750	-2	0	10	+0.006
49.270	11	1.8479	-2	2	7	-0.003	58.659	4	1.5725	-2	4	8	-0.007
49.347	11	1.8452	0	2	10	-0.004	58.727	4	1.5709	3	1	5	-0.013
50.249	1	1.8142	-1	3	9	-0.006	58.893	13	1.5668	2	6	4	+0.011
50.481	9	1.8064	-1	1	10	-0.003	59.043	10	1.5632	1	5	9	+0.000
50.565	10	1.8036	2	2	7	-0.005	59.148	1	1.5607	-1	3	11	+0.002
51.066	1	1.7871	1	3	9	-0.001	59.302	1	1.5570	0	2	12	-0.016
51.198	25	1.7828	-1	5	7	-0.001	59.704	2	1.5475	-3	1	6	+0.001
51.391	11	1.7765	1	1	10	-0.003	59.972	1	1.5412	2	4	8	-0.006
51.829	24	1.7625	1	5	7	+0.000	60.070	3	1.5389	0	4	11	0.008
52.069	2	1.7550	2	0	8	+0.012	60.210	3	1.5357	-1	1	12	-0.001
52.558	22	1.7398	1	7	0	-0.007	60.459	7	1.5300	-2	2	10	+0.000
52.746	8	1.7340	-1	7	1	-0.008	60.887	1	1.5202	2	6	5	+0.006
52.826	7	1.7316	1	7	1	+0.000	61.040	1	1.5168	0	6	9	-0.003
53.373	4	1.7151	-1	7	2	+0.008	61.181	2	1.5136	1	1	12	+0.001
53.558	2	1.7096	1	7	2	-0.001	62.057	7	1.4943	2	2	10	+0.013
53.665	2	1.7065	2	4	6	-0.004	62.150	1	1.4923	-2	4	9	+0.005
54.601	1	1.6794	-1	3	10	-0.007	62.274	1	1.4897	-1	5	10	-0.005
54.727	1	1.6759	1	7	3	+0.005	62.368	5	1.4876	0	8	5	-0.004
55.063	2	1.6664	-2	6	0	+0.005	62.465	1	1.4856	3	3	5	+0.000
55.270	7	1.6607	-1	1	11	-0.007	63.061	1	1.4729	1	5	10	+0.004
55.526	1	1.6536	0	4	10	0.002	63.220	1L	1.4696	2	6	6	+0.035
55.627	8	1.6509	-3	1	4	-0.012	63.908	2	1.4555	-1	3	12	+0.000
55.832	3	1.6453	-2	6	2	-0.045	64.135	1	1.4508	-3	5	1	+0.008
56.021	2	1.6402	-1	7	4	-0.030	64.475	1L	1.4440	0	2	13	+0.009
56.133	1	1.6372	2	6	2	-0.005	64.608	8	1.4414	0	8	6	+0.010
56.208	5	1.6352	1	1	11	-0.004	64.832	6	1.4369	0	4	12	+0.003

and lead in the single-phase state. The solubility limit in similar tungstates was less than 5%.

The ion-conducting properties of the obtained compounds were studied. It was established that $\text{Li}_3\text{Ba}_2\text{Bi}_3(\text{XO}_4)_8$ undergo phase transformations at 441 ($X = \text{Mo}$) and 527 °C ($X = \text{W}$), which, based on the presence of temperature hysteresis in $\lg(\sigma T) - (10^3/T)$ dependences in the heating –

cooling cycle can be interpreted as the first order diffuse phase transitions. After the transition, the conductivity $\text{Li}_3\text{Ba}_2\text{Bi}_3(\text{MoO}_4)_8$ reached values of $3.5 \cdot 10^{-3}$ S/cm (640 °C) at $E_a = 1.0$ eV, $\text{Li}_3\text{Ba}_2\text{Bi}_3(\text{WO}_4)_8$ – $2.7 \cdot 10^{-4}$ S/cm (700 °C) at $E_a = 0.8$ eV. Temperature dependence of electrical conductivity of $\text{Li}_3\text{Ba}_2\text{Bi}_3(\text{MoO}_4)_8$ as an example is shown in Fig. 2. The obtained interdependence

of the active and reactive components of the electrical impedance for this compound (at temperatures before and after the phase transition), typical for ionic conductors with blocking electrodes is shown in Fig. 3.

4. Conclusions

Thus, the first compounds of $\text{Li}_3\text{Ba}_2\text{R}_3(\text{XO}_4)_8$ ($X = \text{Mo}, \text{W}$) family were obtained with the structure of $\text{BaNd}_2(\text{MoO}_4)_4$ (sp. gr. $C2/c$, $Z = 2$),

containing different from the rare earth elements a trivalent metal. The sequence of chemical transformations occurring during the synthesis of ternary tungstate of lithium, barium, bismuth from a stoichiometric mixture of tertiary tungstates and oxides was established. Crystallographic and thermal characteristics of $\text{Li}_3\text{Ba}_2\text{Bi}_3(\text{XO}_4)_8$ ($X = \text{Mo}, \text{W}$) were determined and their ion-conducting properties were studied. It was shown that triple molybdates and tungstates

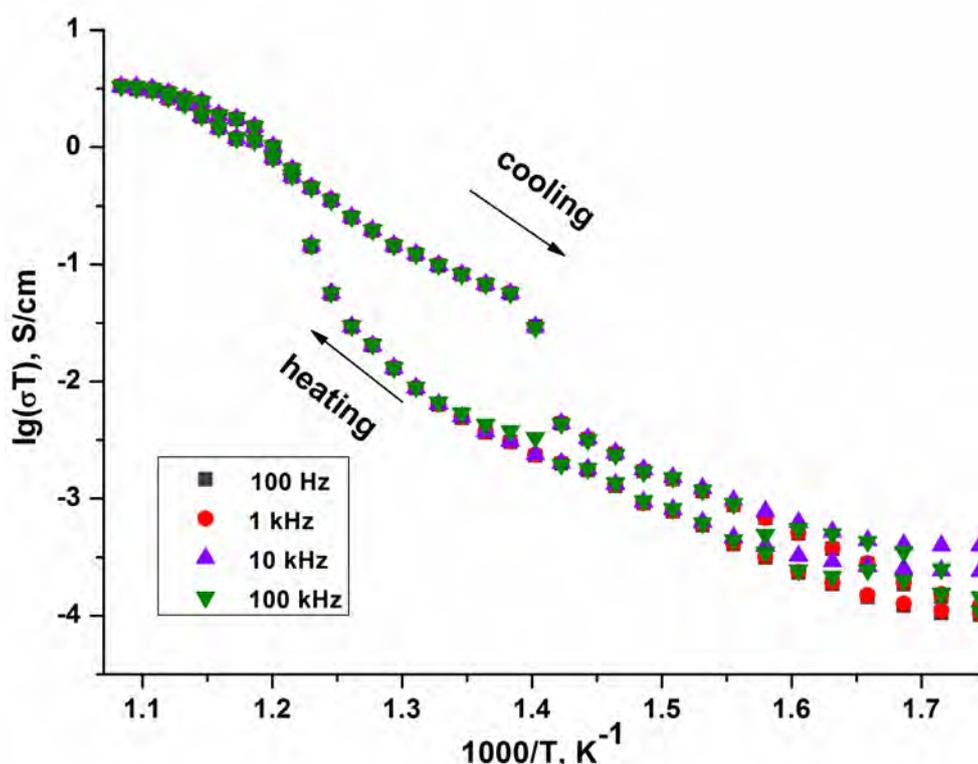


Fig. 2. The temperature dependence of the electrical conductivity for $\text{Li}_3\text{Ba}_2\text{Bi}_3(\text{MoO}_4)_8$

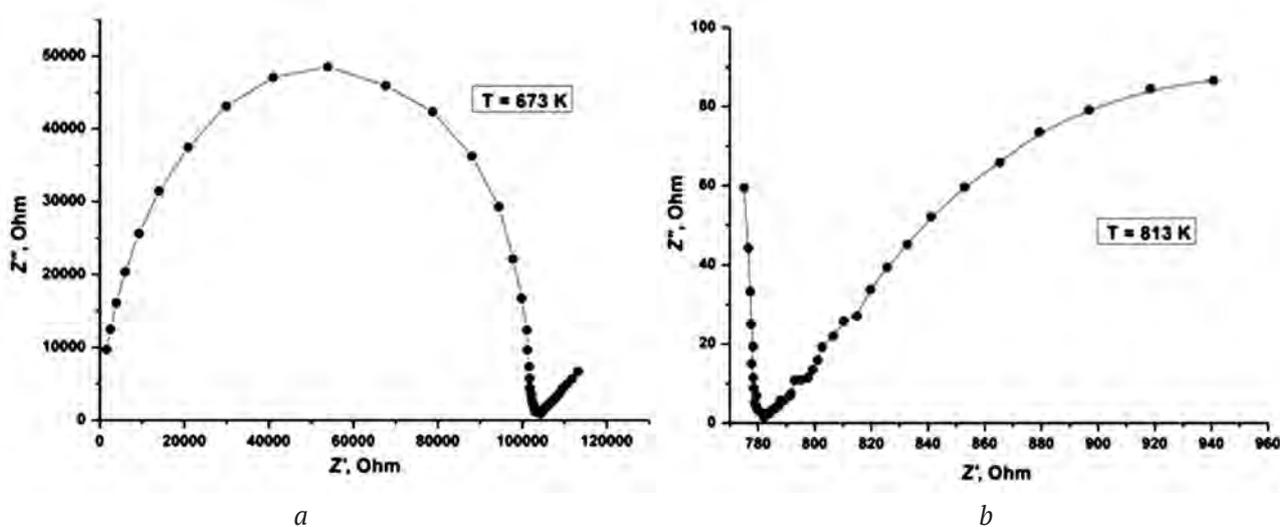


Fig. 3. Nyquist plot for $\text{Li}_3\text{Ba}_2\text{Bi}_3(\text{MoO}_4)_8$ at 673 K (a) and 813 K (b)

$\text{Li}_3\text{A}_2\text{Bi}_3(\text{XO}_4)_8$ ($\text{A} = \text{Ca}, \text{Sr}, \text{Cd}, \text{Pb}; \text{X} = \text{Mo}, \text{W}$) with the structure $\text{BaNd}_2(\text{MoO}_4)_4$ are not formed.

Conflict of interests

The authors declare that they have no known competing financial interests or personal relationships that could have influenced the work reported in this paper.

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